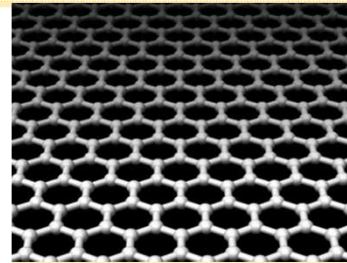


FIG. 2. SEM image of the diamond coating on CuN layer. A diamond crystal (size 5–6 μm) displays a hexagonal morphology.



Transitions de phase dans la croissance de matériaux carbonés

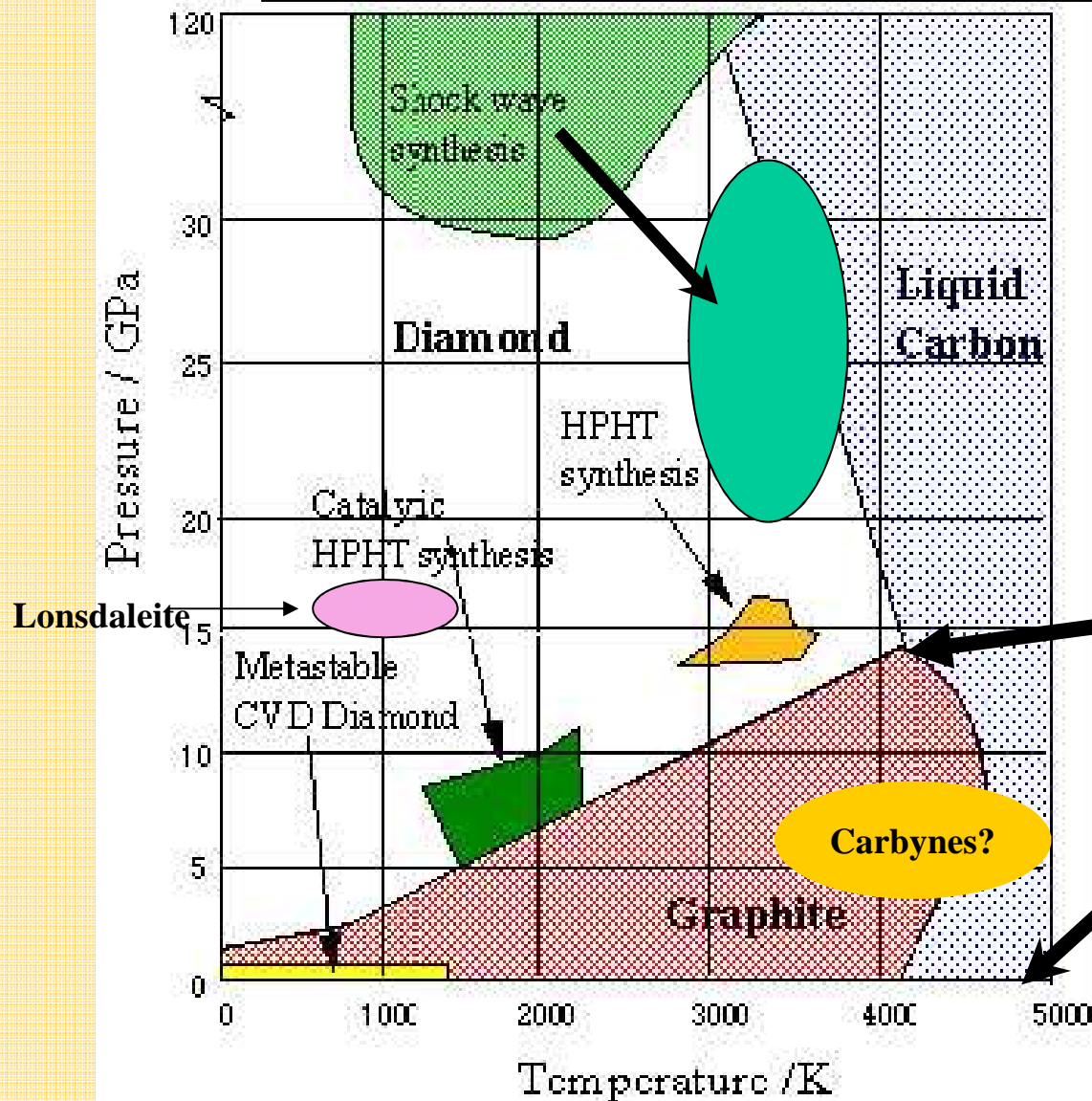
F. Le Normand

*ICube-MAPECV, CNRS/Université de Strasbourg,
STRASBOURG, FRANCE*

Plan

- *Diagramme de phase du carbone*
- *Spectroscopies d'électrons (XPS, AES) appliquées au carbone*
- *Exemples de transition de phase du carbone*
 - * *La nucléation hétérogène du diamant métastable*
 - * *Le graphène sur diamant*

Diagramme de phase du carbone (diagramme de Bundy)



Pour obtenir la transition du
graphite au diamant



Très fortes pressions
Hautes températures

Point triple

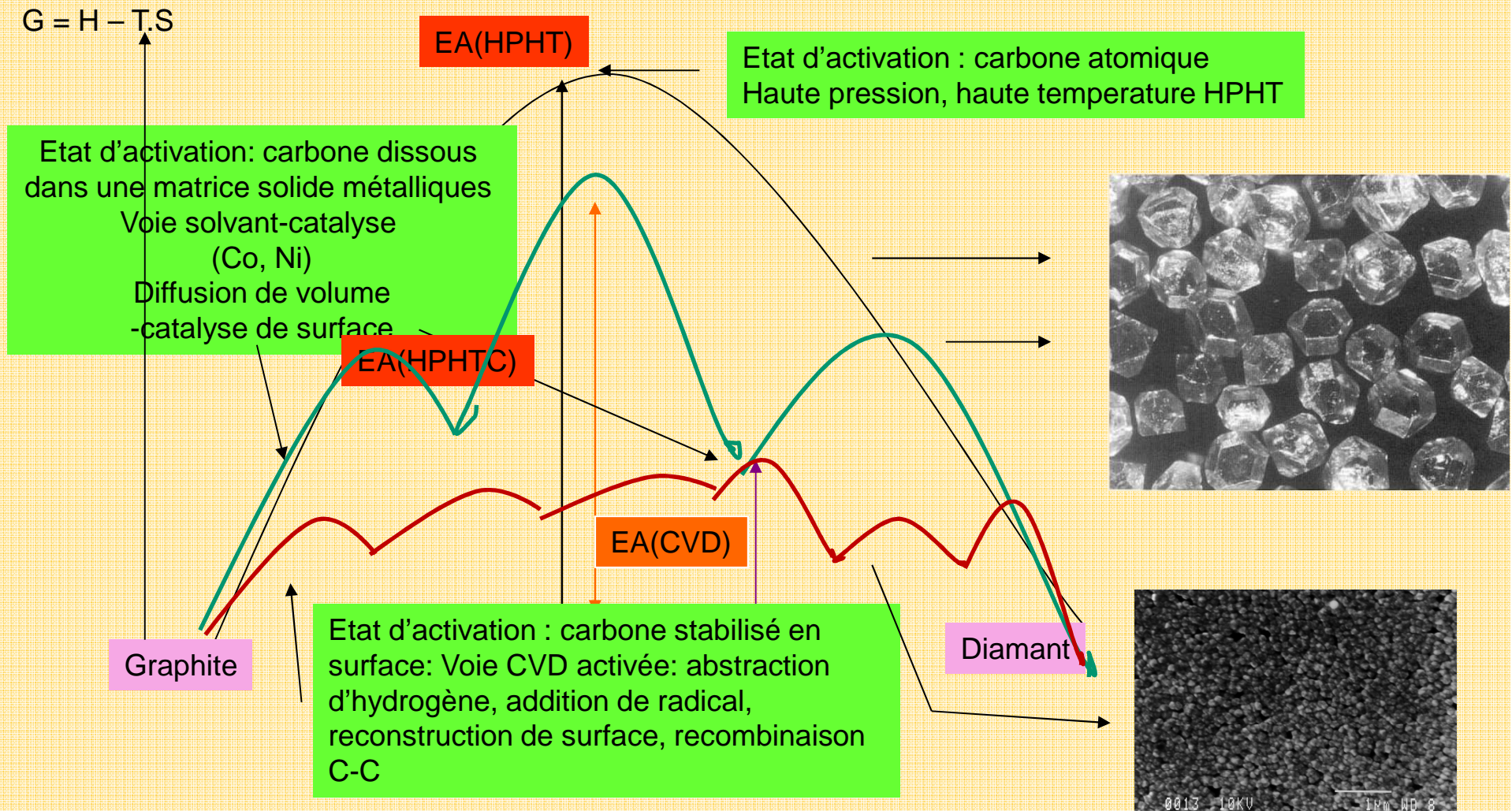
Vaporisation du carbone
4800K; 0.1 GPa

Transition Graphite-Diamant

$$p_{\text{eq/atm}} = 7000 + 27T \quad (\text{for } T > 1200 \text{ K})$$

Bundy's equation

La différence des énergies libres entre diamant et graphite est faible, mais l'énergie d'activation est énorme car il faut rompre toutes les liaisons dans une transformation en phase solide



C1s XPS du carbone

From Egelhoff Jr, *Surface Science Report* 6, (1986) 264

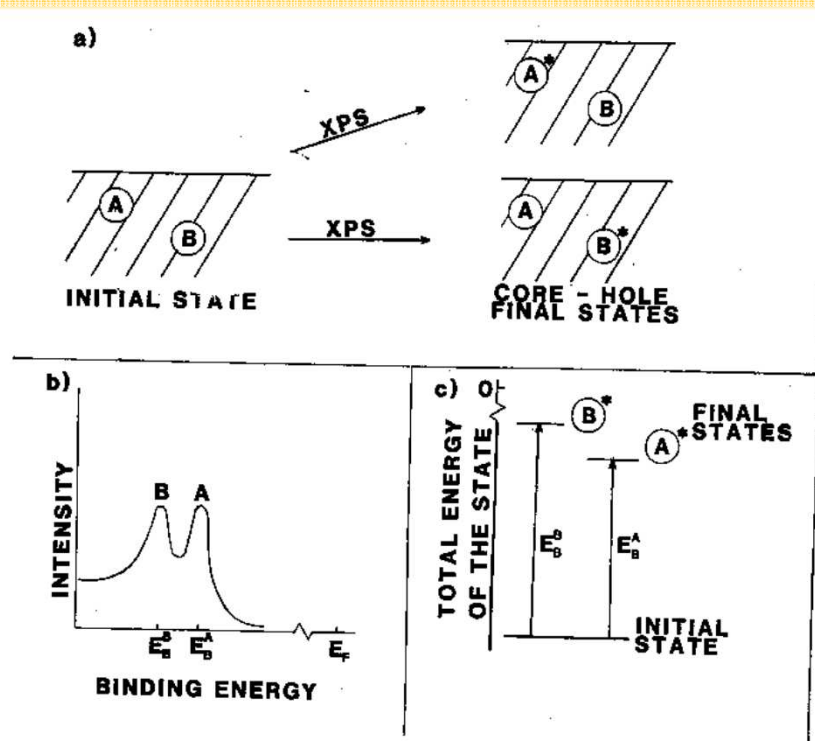


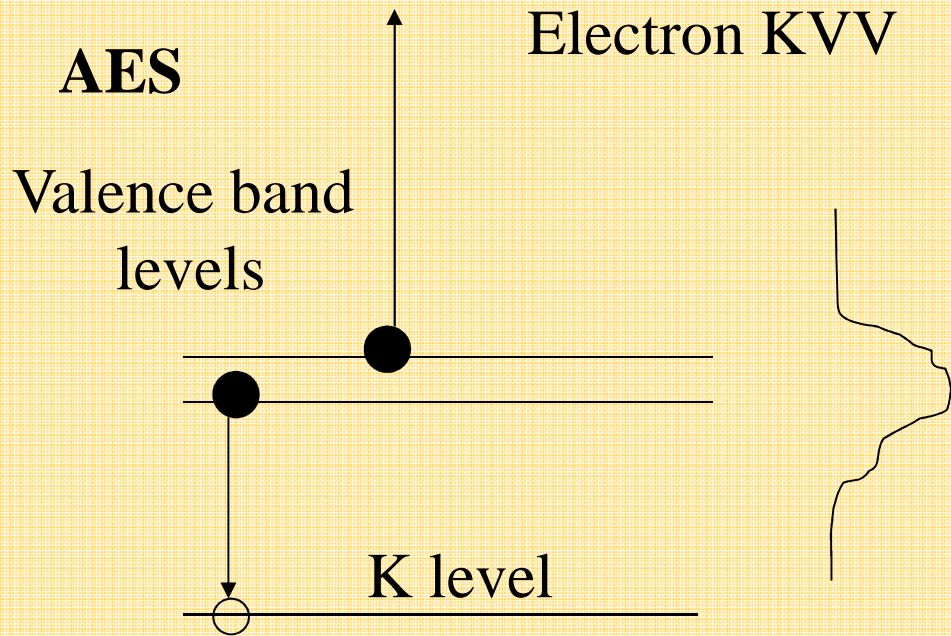
Fig. 2. Illustration of the shift in core-level binding energy for the same element in two different environments, A and B. The initial and final states in a total-energy differences framework are illustrated in (a), the core-level peaks in (b) and total energies of the states in (c). Final states of the type in (c) will be referred to throughout this review with A* described as the state of lower total energy and B* described as the state of higher total energy.

XPS : environ +1 eV entre C-C (graphitique, 284,6eV) et C-C (diamant, 285,5 eV) due à effet d'état final et niveau de Fermi du diamant

Ecrantage du niveau de coeur vacant par électrons π délocalisés > électrons σ très localisés

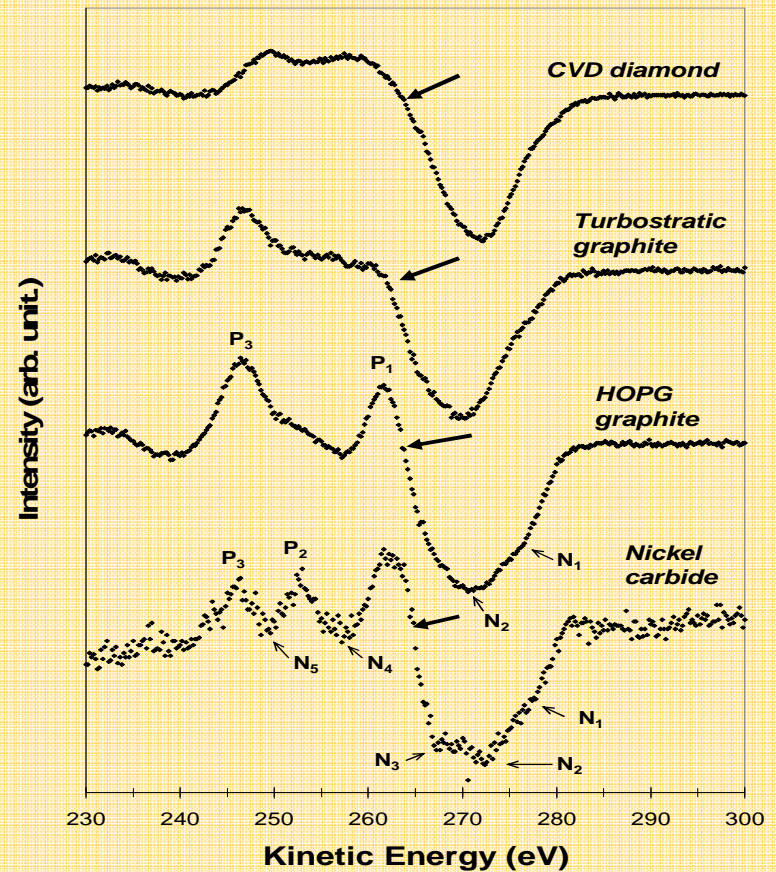
AES du carbone (KVV)

Mode dérivé



Double convolution de la densité des états de valence
 DOS o DOS*

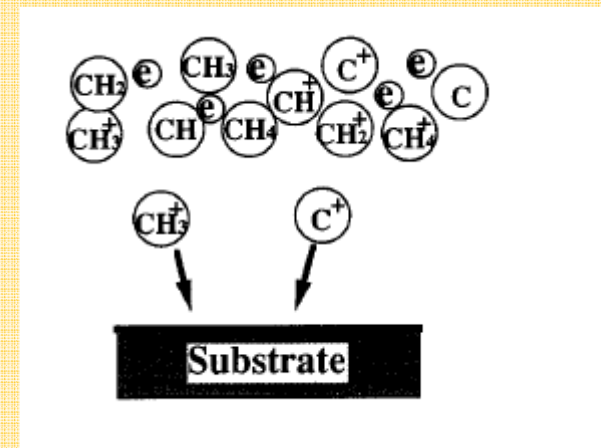
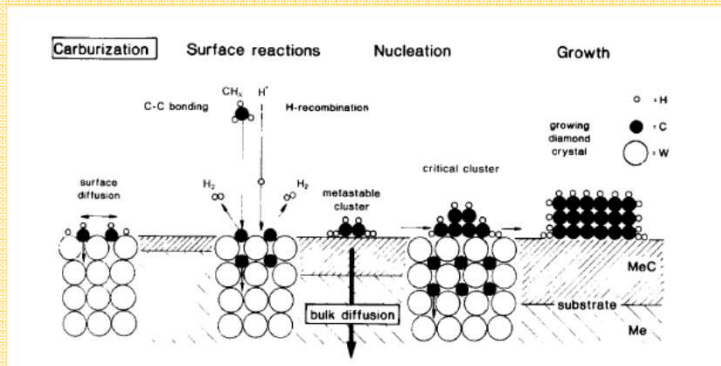
Grande sensibilité à l'état d'hybridation du carbone



Our reference data base

Nucléation hétérogène du diamant CVD (métastable)

Nucléation classique : très faible densité Nucléation assistée par polarisation (BEN-CVD)



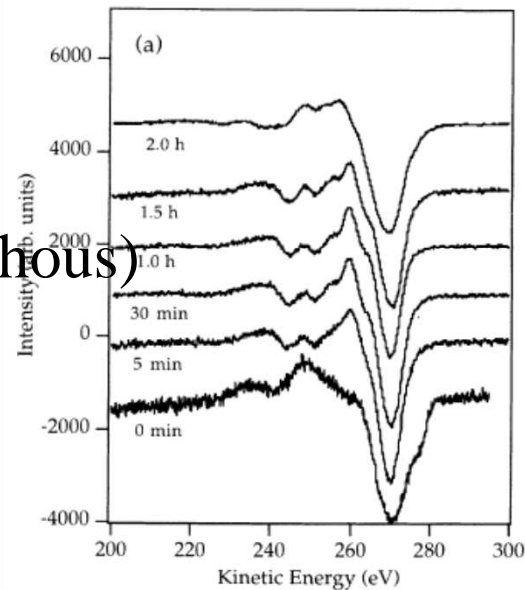
Bias Enhanced Nucleation-Micro-Wave Assisted CVD du diamant sur Si(100)

- Many reports in the literature about the nature of the **diamond precursor**:

- Graphitic fragments parallel or perpendicular to the surface
- Diamond like carbon (amorphous)
- Silicon carbide

Two mechanisms

- Surface nucleation
- Subimplantation



Formation de carbure de silicium

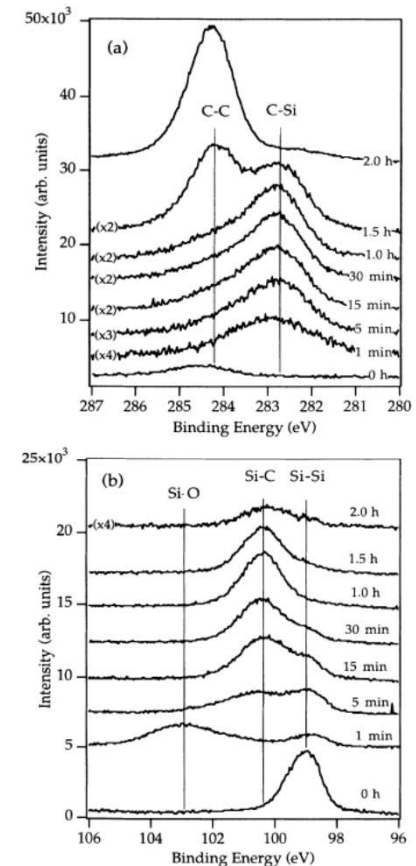
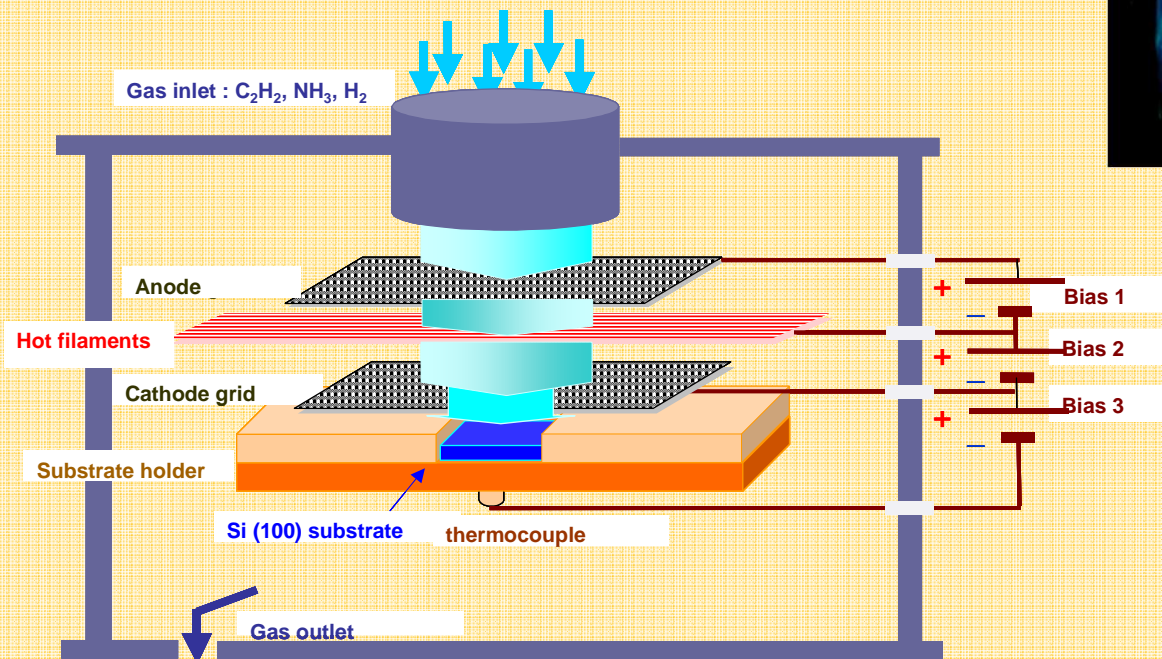
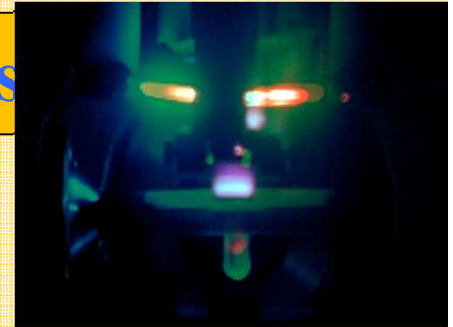


FIG. 5. XPS analysis as a function of bias-pretreatment time: (a) carbon 1s and (b) silicon 2p peak progressions.

Procédé CVD assisté par radicaux et par ions



XPS, AES
chamber



Polarisation between electrodes

ions

Hot filaments

radicals

- Bias 1 + Bias 2 : Plasma control (150-400V)
- Bias 3 : Control of the ions flux onto the substrate (0-60V)
- Nucléation : Bias + HF; Croissance : HF

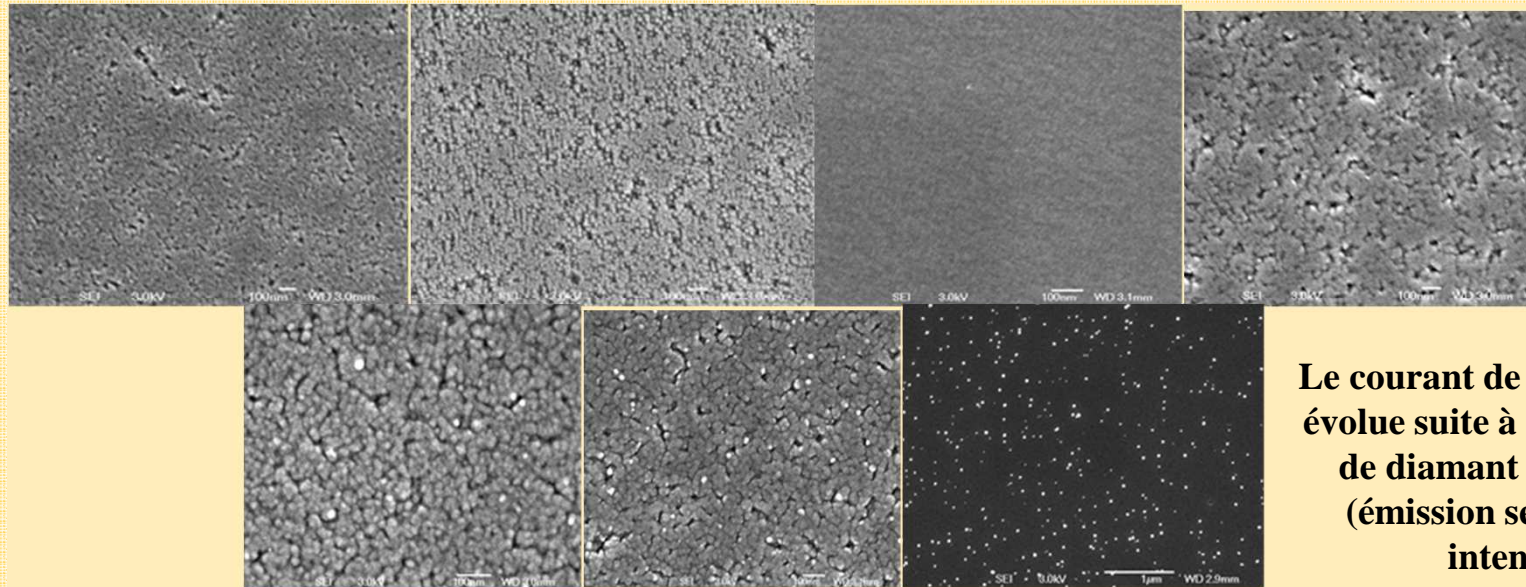
SEM, AES et courant de polarisation Nucleation BIAS -HF CVD on Si(100)

Temps(mn) 2

5

7

9

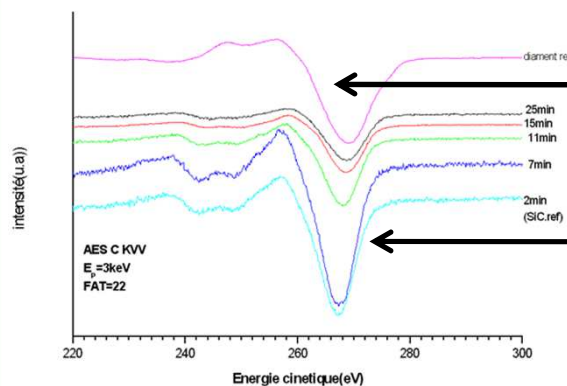


C KVV

11

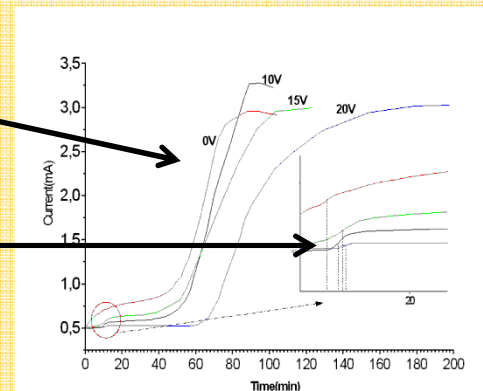
15

25



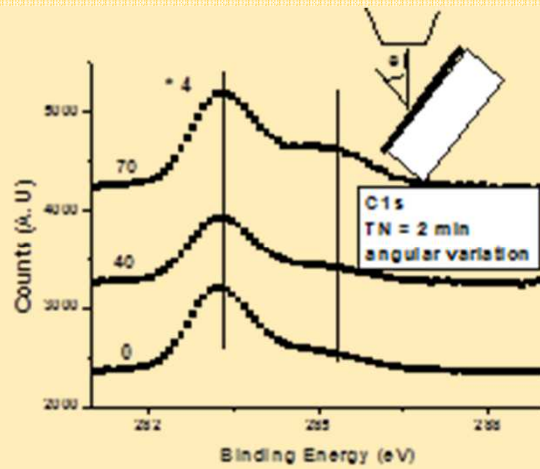
Diamant

SiC

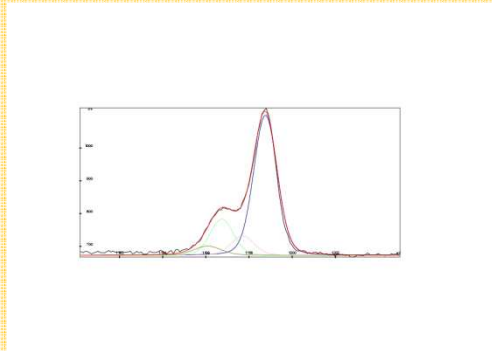


Analyse angulaire XPS C1s et Si 2p Nucleation BIAS –HF CVD on Si(100) ($T = 700^{\circ}\text{C}$; 4% CH_4)

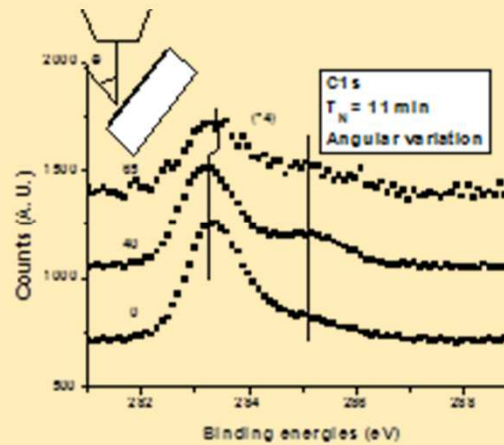
$t = 2 \text{ min}$



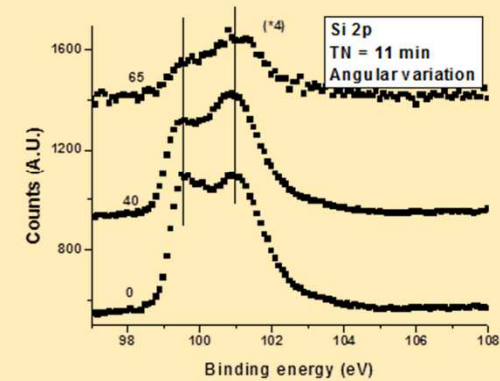
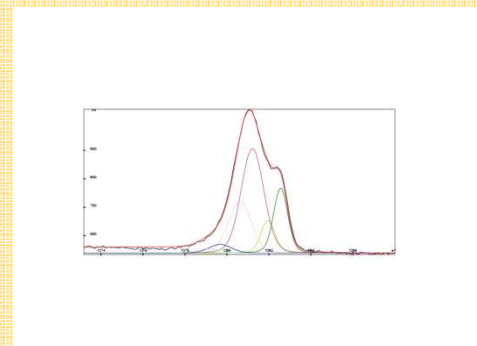
C1s



$t = 11 \text{ min}$



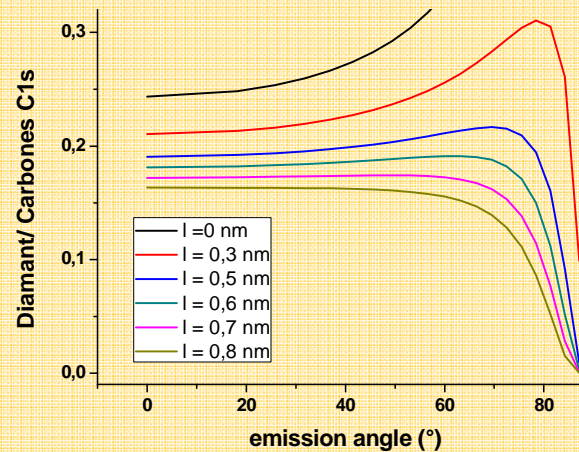
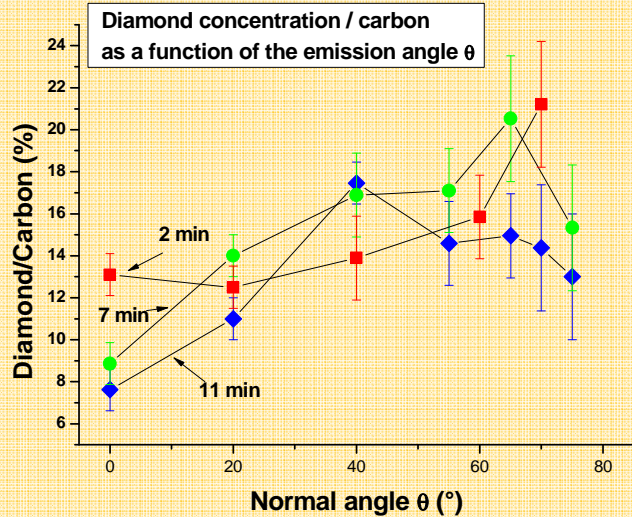
C1s



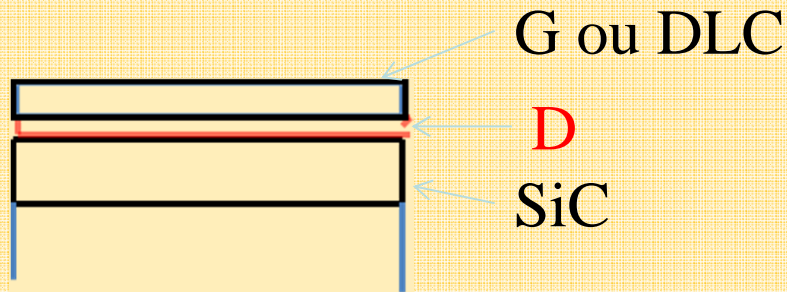
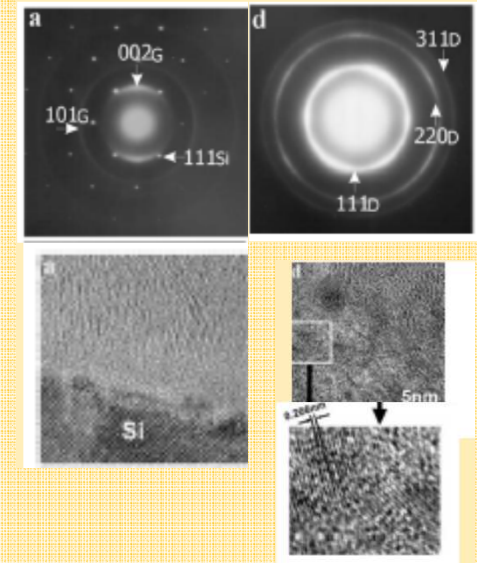
Si 2p

BEN -HF CVD on Si(100)

(T = 700°C; 4% CH4)



Lifschitz et al, Phys Rev Letters, 93 (2004) 056101

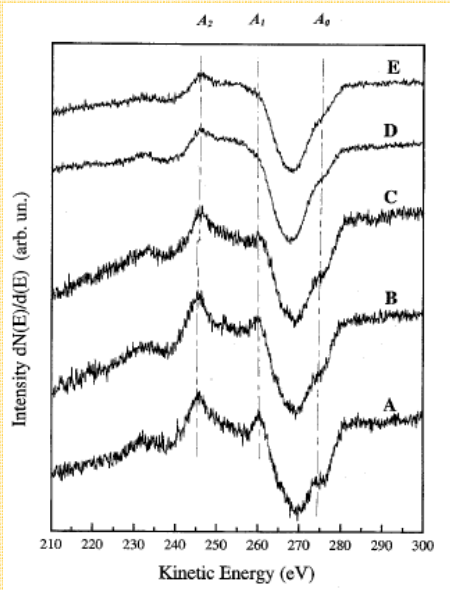


- Mécanisme d'implantation superficielle
- Par contre on ne peut trancher sur la nature du précurseur

$$I_D = 2 * A * n_D * \lambda_{CSi} * \cos\theta * \exp[-L/(\lambda_{CSi} * \cos\theta)] * \text{sh}[c/(2 * \lambda_{CSi} * \cos\theta)]$$

HF CVD sur Cu(111)

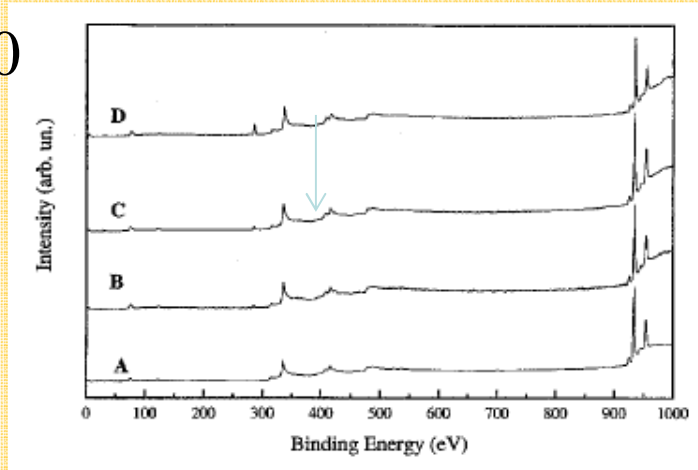
C KVV



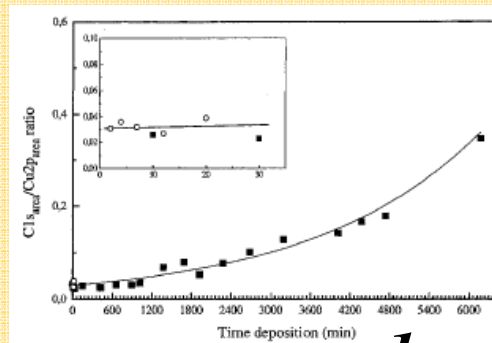
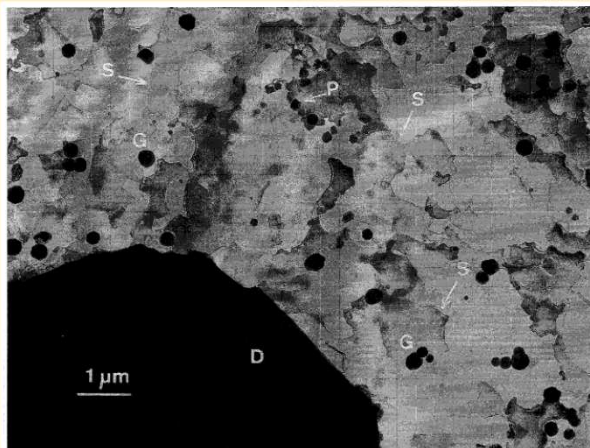
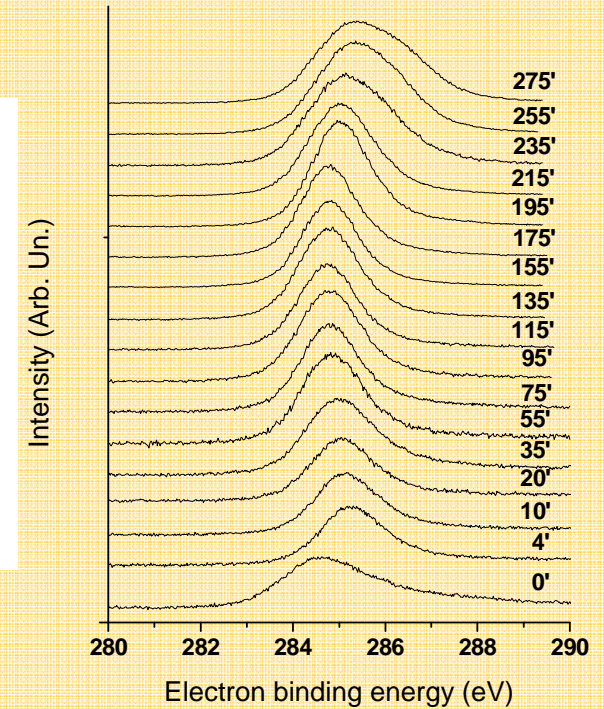
4020
900
100
10
0

L. Constant, et al, Surface Science, 387, (1997), 28-43

XPS wide scan



C 1s

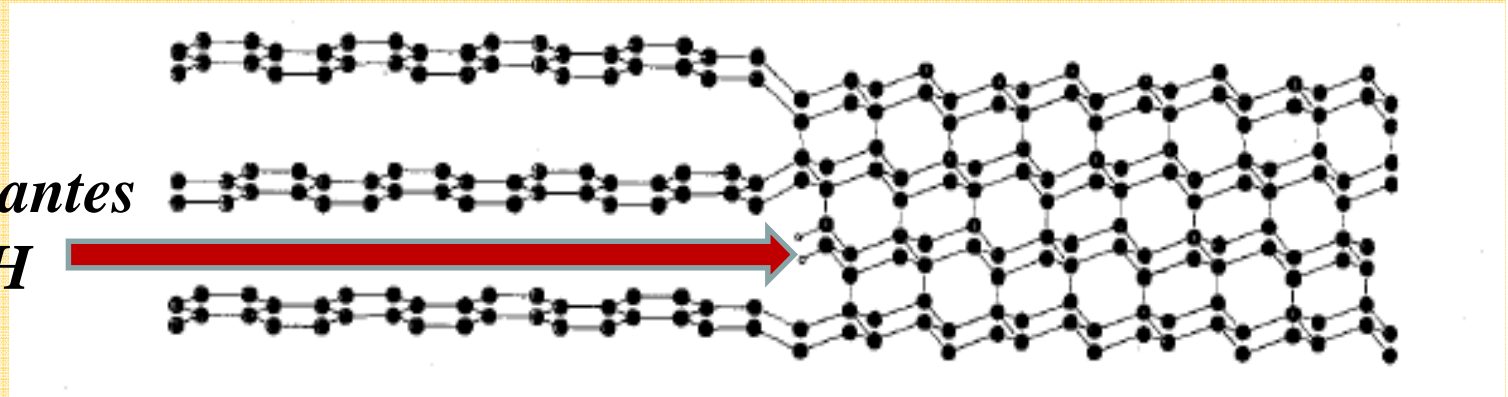


1 monocouche équivalente de graphite (graphène) à $t=0$ min

Croissance de diamant sur cuivre

from the edge of graphitic fragments

*2 liaisons pendantes
comblées par H
radicalaire*



Séquence d'empilement G(00.1)//D(111) : 2//3

Lambrecht W.R.L. et al., Nature, 364, 607, (1993)

Thin graphene layers (TGL) sur Diamond like carbon (DLC)

DLC (ta:C)

- Surface non polaire
- Faible rugosité
- Faible densité de piégeage des charges
- Inerte chimiquement
- Procédé simple
- Stable thermiquement

**Mais transfert de
graphène (CVD cuivre)**

Graphène sur DLC (transistor RF)

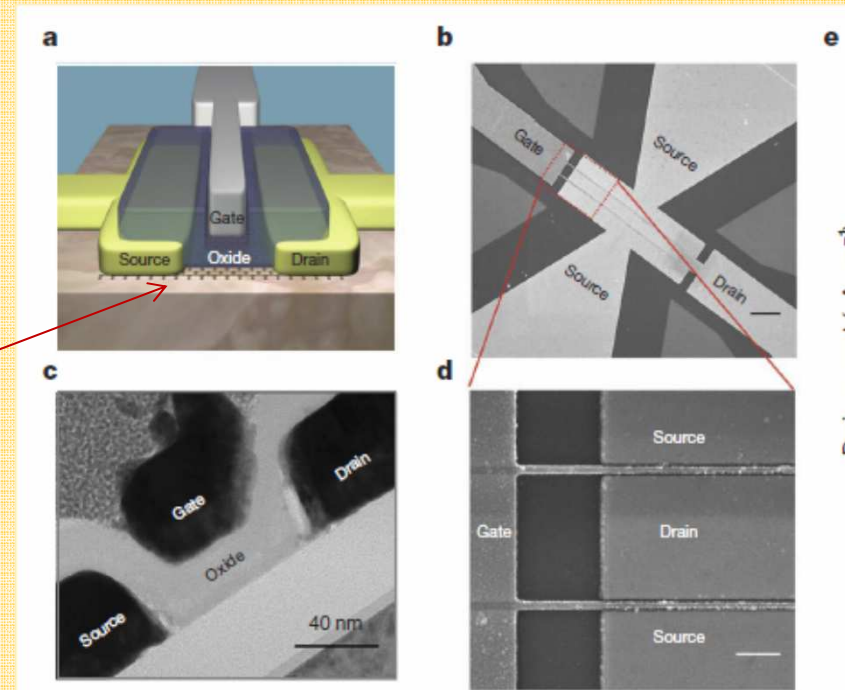
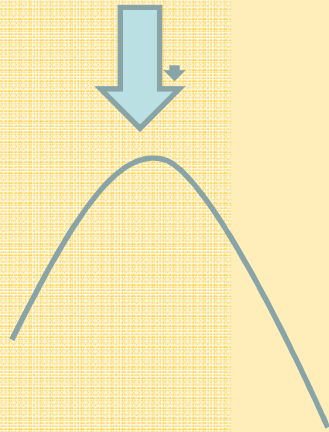


Figure 1 | Fabrication and output characteristics for graphene r.f. transistors. a, Schematic view of a top-gated graphene r.f. transistor on DLC substrate. b, SEM image of a typical top-gated dual-channel r.f. device. Scale bar, 3 μm . c, Cross-section TEM image of a graphene transistor with a gate

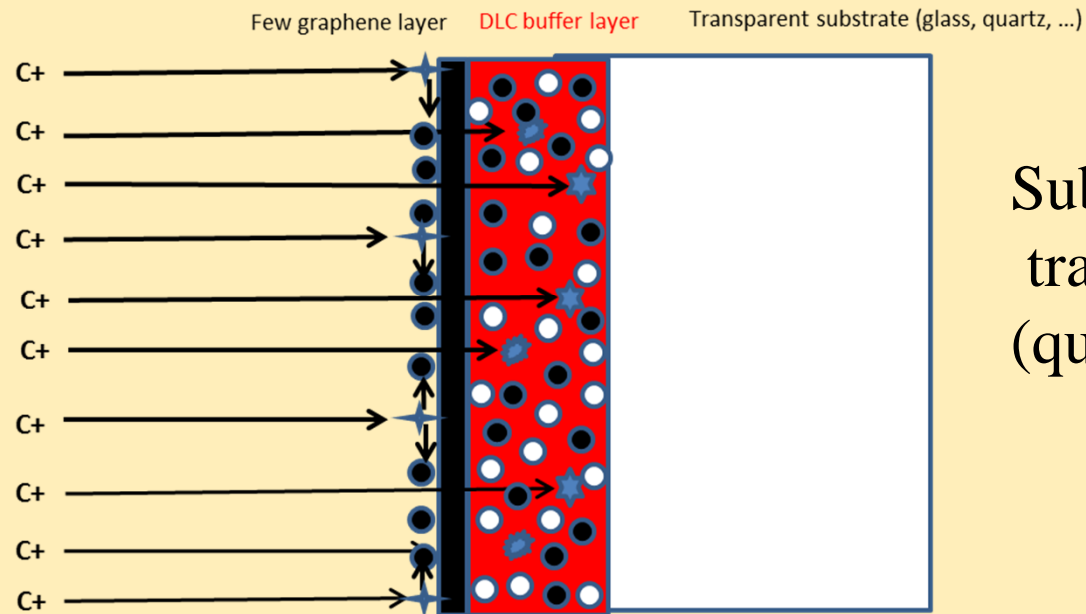
Croissance de DLC

A scheme of continuous growth of few graphene layers on top of DLC growing film by pulse laser ablation of carbon

$\sim 100 \text{ eV}$



Distribution
d'énergie des
ions



Substrat
transparent
(quartz, verre,)

- ★ Surface impinging and diffusion at low C^+ energy with sp^2 carbon formation
- ★ Subplantation with stress (without relaxation) at medium C^+ energy with sp^3 carbon formation
- ★ Subplantation without stress (with relaxation) at high C^+ energy with sp^2 carbon formation

D'après D.R. MacKenzie et al., *Phys rev. Letters*, 67 (1991) 773

Etude surface DLC

S. Rey et al, Applied Physics A, 71 (2000) 433-439.

C1s core level XPS on DLC films with variable fluences after a thermal annealing at 923K (left); angular resolved for a DLC film at 10 J/cm² post-annealed at 923K (right)

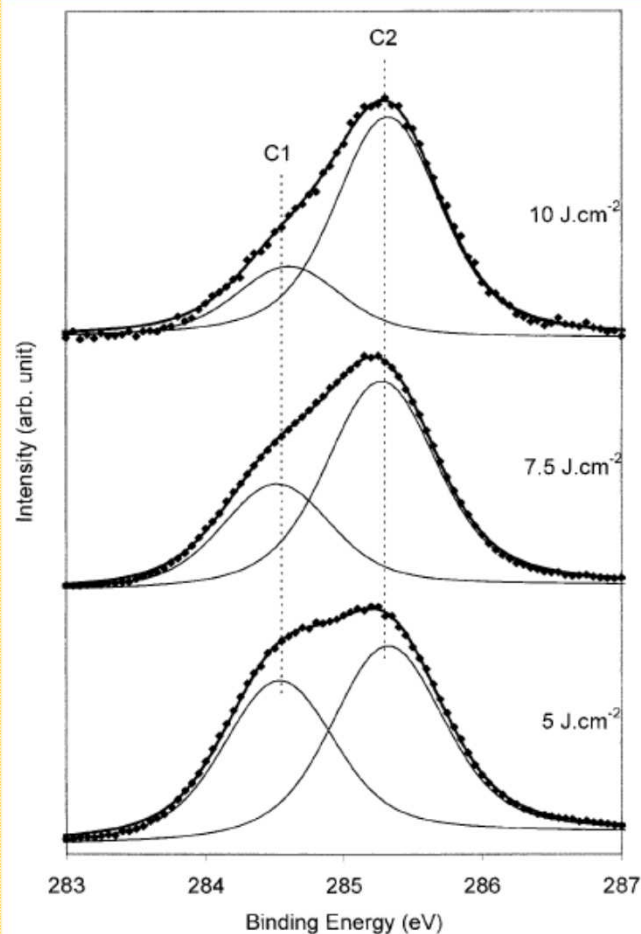


Fig. 1. XPS C 1s spectra obtained on ns($X-923$)-type samples as a function of the laser fluence X

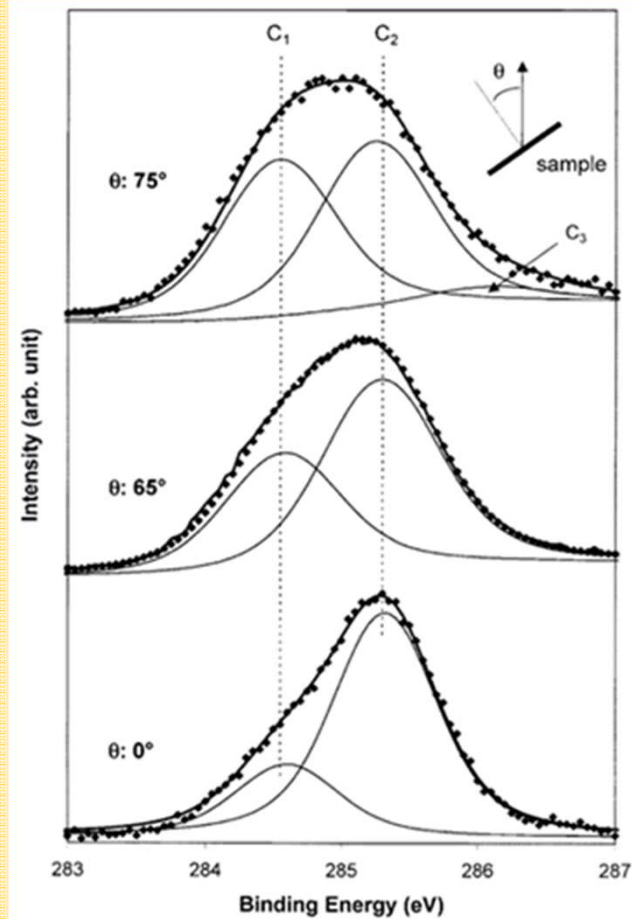
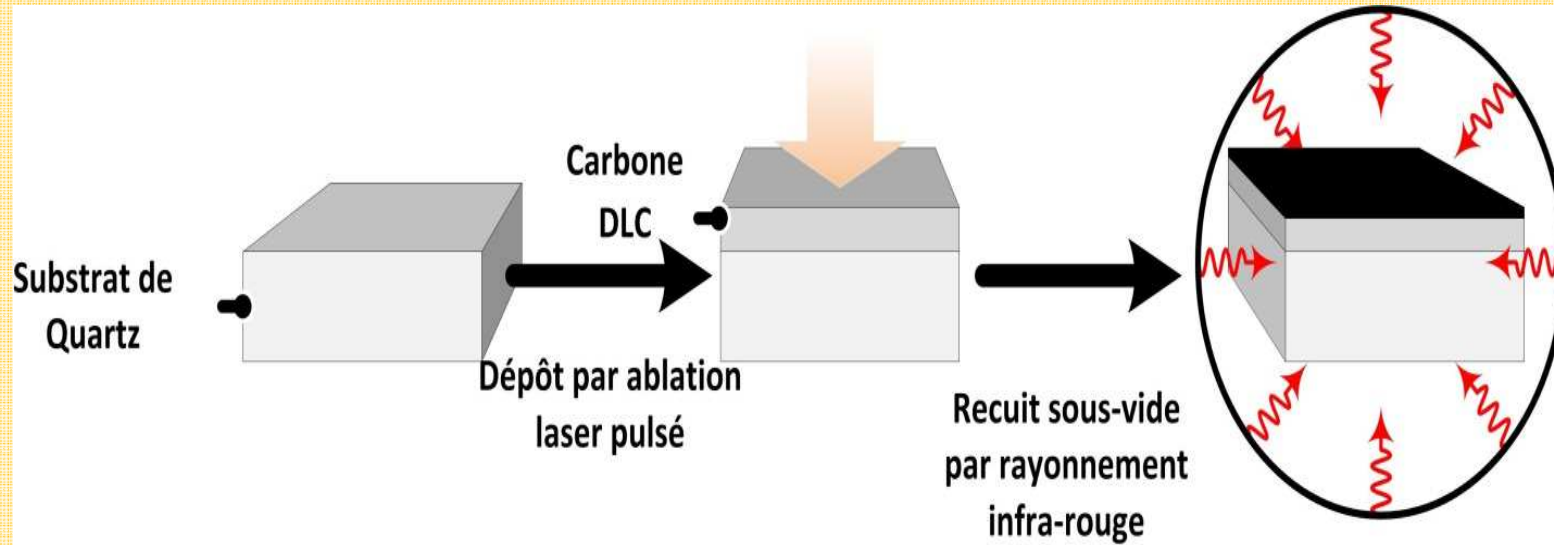
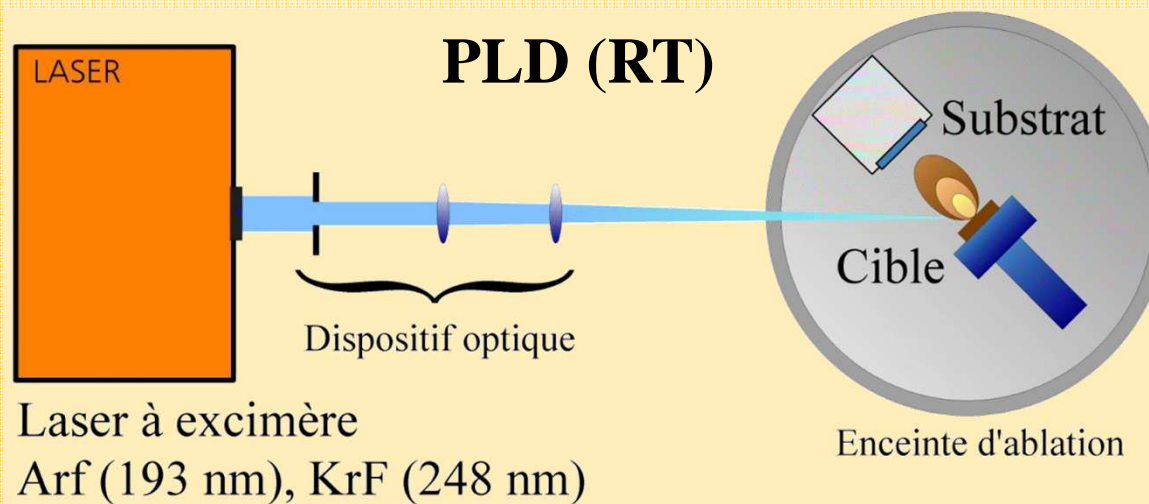


Fig. 2. Angular XPS analysis of the C 1s core level obtained on the ns(10.0-923) sample. θ is the angle between the normal of the sample surface and the direction of the emitted electrons

Procédure d'élaboration de Graphene/DLC/Quartz)



ou traitement laser

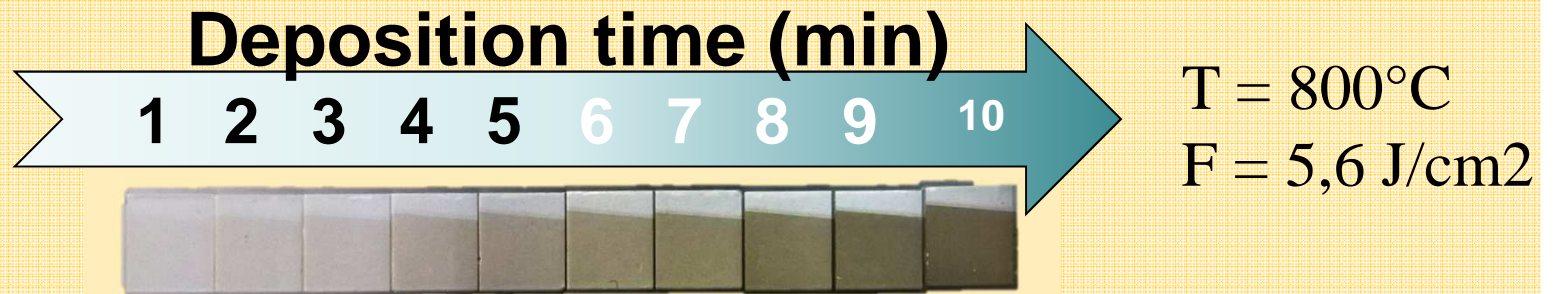


Conditions de Dépôts et traitements thermiques

$$1 \text{ J/cm}^2 < \text{Fluence} < 10 \text{ J/cm}^2$$

$1 \text{ min} < t < 10 \text{ min}$ soit épaisseur DLC entre 3 et 30 nm

$$600 < T_{\text{recuit}} < 1100^\circ\text{C}$$



5 cm

Très bonne uniformité
Rugosité < 1 nm

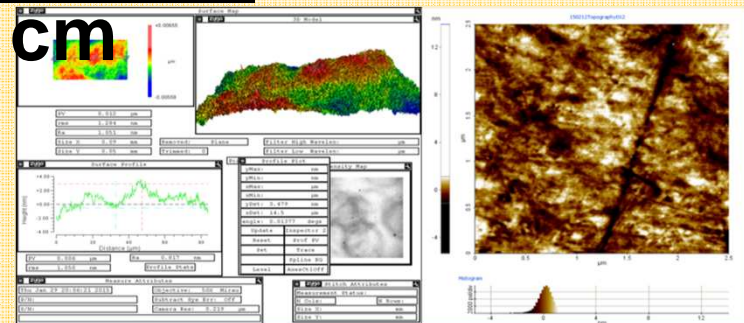
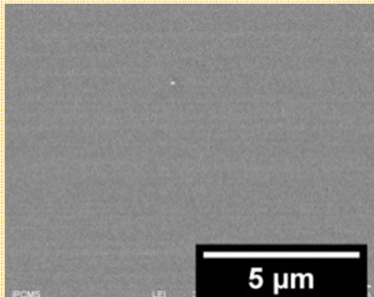
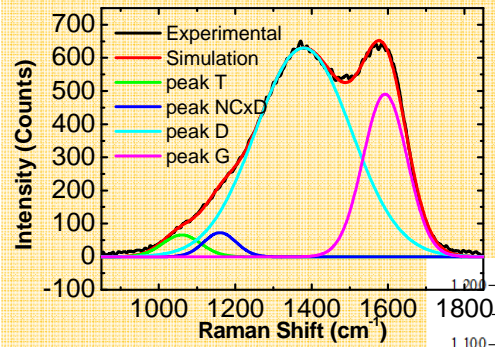


Figure Erreur ! Il n'y a pas de texte répondant à ce style dans ce document.1 On the left Interferometry measurements, on the right the AFM image in Non-contact mode of DLC_quartz_10

Spectroscopie Raman : DLC sur Quartz

apres post-traitement thermique UHV

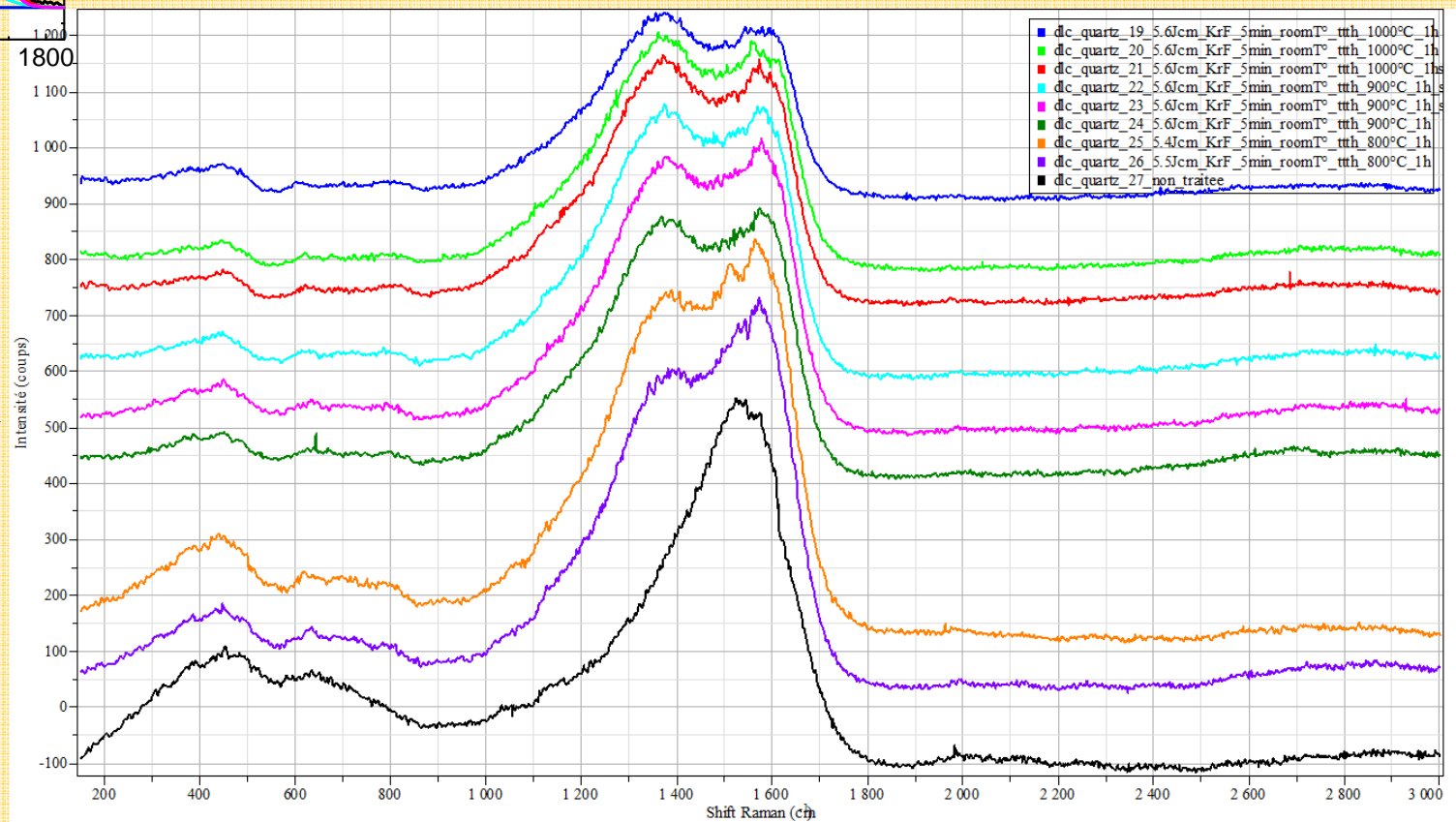


600°C < T < 1000°C

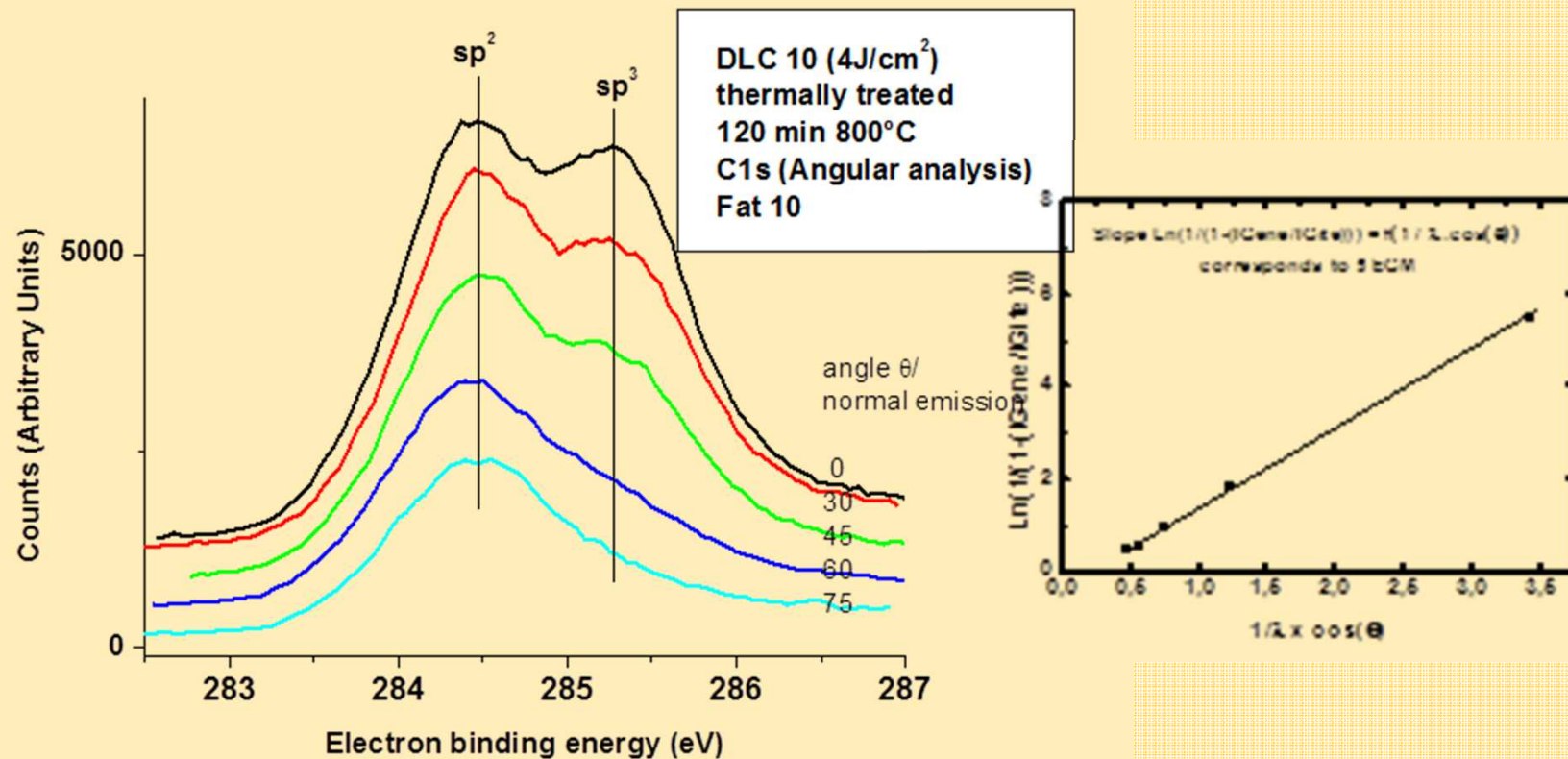
D G

2D

Raman
non pertinent



XPS : a proof of the occurrence of a graphene layer on top of DLC

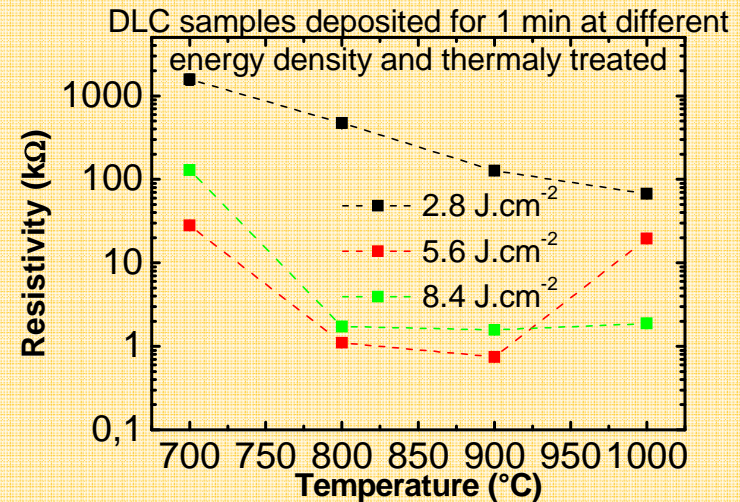
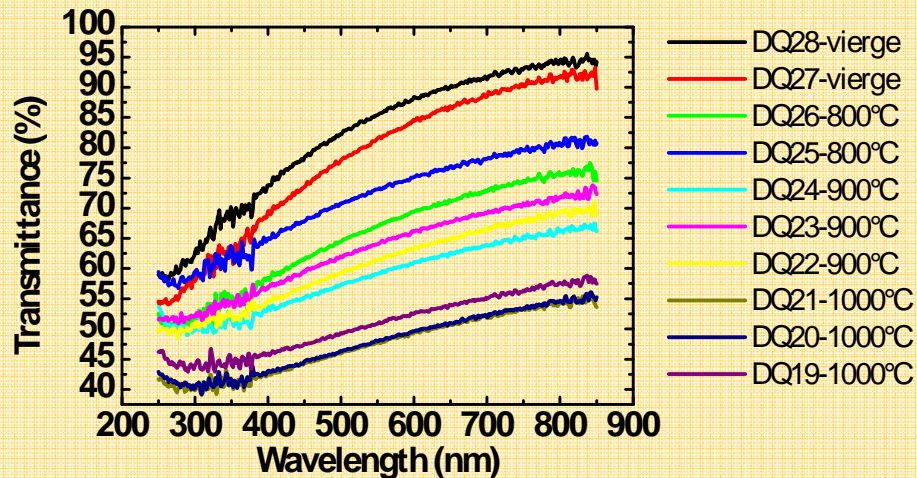


Mesure de l'épaisseur de la couche TLG

$$e = \lambda_C * \cos\theta * \text{Ln}\{1 / [1 - (I_{\text{Gene}} / I_{\text{Gite}})]\}$$

With $\lambda_C = 2,1 \text{ nm}$ $e \sim 1,0 \text{ nm}$

Transmittance and sheet conductivity



Avec transmission environ 90% et
resistivités surfaciques $\sim 1 k\Omega/cm^2$ la figure
de mérite des électrodes transparentes
« conductivity of transparency »
est du même ordre de grandeur que ITO



Diapositive 22

2ndF1

Fitsum AWEKE; 18/05/2015

Conclusions

- Le dépôt de carbone par HF CVD conduit à des transformations de phase du carbone sur silicium et cuivre
- Le mécanisme de nucléation aussi bien que la nature des précurseurs varient suivant les conditions
- Les spectroscopies de surface apportent des informations importantes
- Production très simple de “thin layers graphenic layers” sur DLC par transformation de surface
- avec de bonnes performances pour des électrodes transparentes
- Quelle est la véritable nature de ces couches “graphéniques”?

Remerciements

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J. Faerber (IPCMS/GSI, Strasbourg)

J. Hommet (IPCMS/GSI, Strasbourg)